



Quantification of PF₅ and POF₃ from Side Reactions of LiPF₆ in Li-Ion Batteries

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The electrochemical oxidation of LiPF₆-based electrolytes is reported to generate POF₃ gas. In order to enable a quantitative analysis of the LiPF₆ decomposition reactions, we aimed to establish calibration factors for POF₃ and PF₅ in on-line electrochemical mass spectrometry (OEMS). Thermal decomposition of dry LiPF₆ is expected to yield PF₅, but instead all PF₅ is detected as POF₃ in our OEMS setup, rendering a differentiation of the two gases impossible and presenting an artefact which likely occurs with most on-line mass spectrometry systems due to the high reactivity of PF₅. However, we can still determine a cumulative calibration factor for POF₃ + PF₅ (referred to as “POF₃”), which is then used to investigate the evolution of gases during the oxidation of an EC/LiPF₆ electrolyte on a carbon black electrode. Mechanistic experiments with protons or water added to EC/LiPF₆ electrolyte show that protons trigger the formation of PF₅, while the kinetics for the hydrolysis of LiPF₆ with water at room temperature are too slow to be detectable. These findings let us conclude that the oxidation of EC generates highly acidic species, which cause the decomposition of PF₆⁻ to PF₅ and HF; the PF₅ is then detected as POF₃ in the OEMS.

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Lithium hexafluorophosphate (LiPF₆) is commonly used as a salt in Li-ion battery electrolyte solutions due to its high ionic conductivity and passivating properties toward the aluminum cathode current collector.¹ However, its thermal instability limits the operating temperature range for Li-ion batteries to < 60°C, whereas its reactivity with water calls for dry-room manufacturing of cells. The corresponding primary decomposition pathways are: i) the thermal dissociation of LiPF₆, leading to PF₅ and LiF; and, ii) the hydrolysis of LiPF₆ or PF₅, resulting in the formation of HF and POF₃.

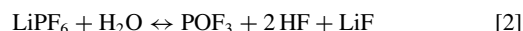
A quantitative thermal dissociation of dry LiPF₆ occurs between 100 and 200°C, depending on the experimental conditions (i.e., sealed or open containers,² sample size),^{3,4} according to the following equilibrium:^{2–5}



Common Li-ion battery electrolytes, namely anhydrous solutions of LiPF₆ in a mixture of ethylene carbonate (EC) and dialkyl carbonates, show negligible thermal aging up to 60°C.^{6,7} If storage temperatures of 80°C are exceeded, the electrolyte solution darkens and forms large amounts of gas within days.^{2,8–11} Next to PF₅,² also alkyl fluorophosphates,^{2,7,9–14} oligomers of the carbonate solvents,^{8,10} fluorinated hydrocarbons,^{2,10,11} POF₃,¹³ and HF^{13,14} have been found in thermally aged (80–100°C) electrolyte solutions, indicating that not only LiPF₆, but also the organic carbonate solvent itself is decomposed. However, the observed quantities differ significantly across studies. Campion et al.^{9,10} and Guillot et al.¹¹ showed that impurities like alcohols and water increase the amount of side products at high temperatures. Furthermore, electrolytes stored in glass vials between 60–85°C show about 100 times more decomposition products compared to the same electrolytes stored in polymer or aluminum containers,^{7,14} as the etching of SiO₂ by HF generates water, leading to a self-accelerating decomposition of the electrolyte solution.^{15,16} Hence, it is clear that the purity of the electrolyte and the testing conditions substantially affect the thermal stability of LiPF₆-based electrolytes.

The hydrolysis of LiPF₆ with trace water in Li-ion battery electrolyte solutions occurs already at room temperature, although a complete conversion needs days to weeks.^{17–21} The main product of LiPF₆

hydrolysis with H₂O is HF and POF₃ (see Equation 2);^{21,22} the latter is further hydrolyzed to HPO₂F₂ according to Equation 3.^{18,19,23} The first reaction leads to a 2:1 molar ratio of formed HF to consumed water, as recently found by Strmcnik et al.²¹ At temperatures below 60°C, no reactions between the organic electrolyte solvent with PO₂F₂⁻ or HF generated from LiPF₆ hydrolysis have been reported.^{19,23}



Besides these two major decomposition pathways, LiPF₆ also affects the stability of the electrolyte solvents at high potentials. Density functional theory (DFT) calculations indicate that the oxidation potential of EC-PF₆⁻ complexes is lower than for isolated EC, and that HF and PF₅ can be formed from the oxidized complexes at room temperature.^{24,25} Tebbe et al.²⁶ suggested that coordination to PF₅ reduces the activation energy for EC dimer formation or ring opening. In fact, PF₅ has been reported to slowly react with carbonate electrolytes already at room temperature,⁸ and alkyl fluorophosphates (the characteristic products of PF₅ with organic carbonates) have been found in cells cycled to potentials above 5.2 V vs. Li/Li⁺.^{12,27} Still, PF₅ has not been observed directly during electrochemical electrolyte oxidation so far. Instead, POF₃ has been detected at high potentials in LiPF₆-based electrolytes,^{28–33} where it was ascribed to a reaction of LiPF₆ or PF₅ with water or other reactive oxygen-containing species formed during the oxidation of the electrolyte solvent. While this is plausible for experiments where water or reactive oxygen species are likely to be formed, i.e., in the presence of oxygen-releasing cathode materials,^{28,29,34} the quantitative formation of POF₃ during electrolyte oxidation on inert materials which do not release oxygen upon charge like high-voltage spinel or carbon black still lacks fundamental understanding.^{31–36}

To understand the origin of PF₅ and POF₃ at high potentials in LiPF₆-based electrolytes, a quantification of the evolved amounts of these gases is essential. Therefore, this work aims at quantifying PF₅ and POF₃ by on-line electrochemical mass spectrometry (OEMS). As a first step, we investigate the decomposition of dry and wet LiPF₆ by thermogravimetric analysis coupled with mass spectrometry (TGA-MS). We then thermally decompose LiPF₆ in a modified OEMS cell setup that can be heated up to 250°C. As dry LiPF₆ dissociates completely to LiF and PF₅ at temperatures >200°C according to our TGA-MS analysis (consistent with the literature),^{2–5} we thus can now

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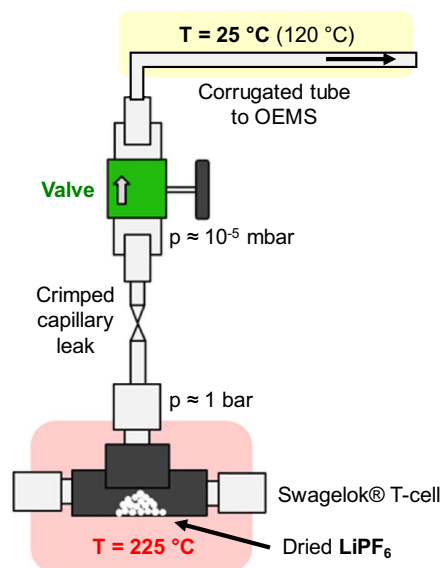


Figure 1. Modified OEMS cell setup used for the thermal decomposition of LiPF_6 consisting of a Swagelok T-cell fitted to a crimped capillary tube, which is connected to the mass spectrometer as described in Ref. 37. The T-cell is then heated to 225°C (red shaded area). The corrugated tube connecting the capillary and the OEMS (yellow shaded area) is normally kept at ambient temperature (i.e., 25°C); for the experiment shown in Figure 4c, the corrugated tube was heated to 120°C for 48 h prior to the experiment.

correlate the ion currents at different mass channels to the gas concentration in the cell. The absence of PF_5 signals due to its high reactivity with OEMS system components ultimately provides an explanation why PF_5 so far has never been detected in on-line mass spectrometric measurements on lithium ion battery cells. Lastly, we investigate the oxidative stability of an EC/ LiPF_6 electrolyte on a carbon black working electrode, thereby focusing on the POF_3 quantification, and resolve the origin of what appears as “ POF_3 ” at high potentials in on-line mass spectrometry measurements on Li-ion batteries.

Experimental

Thermogravimetric analysis coupled with mass spectrometry.—

Thermogravimetric analysis of lithium hexafluorophosphate (LiPF_6 , BASF SE, Germany) was conducted with a Mettler Toledo TGA coupled to a mass spectrometer (Thermostar TGA-MS, Pfeiffer Vacuum, Germany). Prior to the experiments, the LiPF_6 was dried at 70°C for 18 h under dynamic vacuum in a glass oven (Buechi, Switzerland) without exposure to air after the drying step. Then, ~150 mg LiPF_6 were filled into a sapphire crucible inside an Ar-filled glove box (MBraun, Germany) and transferring the crucible into the TGA chamber. For the “dry” experiment, the sample was heated at 10 K/min to 120°C and then held for 1 h at 120°C in a dry argon flow (5.0 purity, 60 mL/min) to remove any traces of H_2O . The sample was then ramped up to 350°C at 10 K/min, where the temperature was held for another 10 min to ensure a complete decomposition of LiPF_6 . For the “wet” experiment, the argon gas (60 mL/min) was saturated with water at room temperature and purged over the LiPF_6 sample. After resting for 1 hour at room temperature under the wet argon flow, the sample was heated at a rate of 10 K/min to 350°C, where the temperature was again kept constant for 10 min.

Modified on-line electrochemical mass spectrometry setup.—To measure the decomposition of LiPF_6 in our on-line electrochemical mass spectrometer (OEMS),³⁷ we designed a modified cell setup (see Figure 1). As pre-tests showed that typical polymer sealings were either not tight at high temperatures or not stable against the evolving PF_5 gas, we used a Swagelok T-fitting with two metal end

caps. The third opening of the T-cell was equipped with the crimped capillary leak that connects the T-fitting to the mass spectrometer (this is also part of our regular OEMS system and has been described previously).³⁷ All cell parts were dried at 70°C under dynamic vacuum prior to assembly. The modified cell was equipped with a thermocouple to monitor the cell temperature and was wrapped with a heating cord (Horst GmbH, Germany); the entire assembly was thermally insulated by fiber cloth (see red shaded area in Figure 1). Tightness of the modified Ar-filled cell was tested in pre-runs of the actual experiments (i.e., holding the cell at temperatures >200°C during several hours), validating that the mass traces $m/z = 28$ (N_2) and $m/z = 32$ (O_2) remained negligible. For the three here shown LiPF_6 decomposition runs, 0.26, 0.96, or 1.08 mg LiPF_6 (± 0.04 mg) was weighed into a TGA crucible inside an Ar-filled glove box. The crucible was then transferred into the modified OEMS cell. The cell was closed inside the Ar-filled glove box and connected to the OEMS. After a rest period of 40 min at 25°C, the temperature was increased to 225°C and held there for ~4 h. The m/z signals were evaluated by dividing the ion current of the respective channel by the ion current of the ^{36}Ar isotope and subtracting the background during the initial rest period at 25°C.

On-line electrochemical mass spectrometry.—Hydrolysis and electrochemical oxidation of 1.5 M LiPF_6 in ethylene carbonate (EC, BASF SE, Germany) was investigated in our regular OEMS cell setup. Note that this system is unfortunately not able to trace HF, as the mass signal for hydrofluoric acid ($m/z = 20$) superposes with the signal of the ^{20}Ar isotope. For hydrolysis experiments, 5000 ppm H_2O or 5000 ppm methanesulfonic acid (99.5%, Sigma-Aldrich, USA) were added to the EC/ LiPF_6 electrolyte. After stirring for 30 s, 240 μL of the electrolyte were transferred into an empty OEMS cell (without any electrodes or separators present). The cell was then placed into a temperature chamber set to 10°C and connected to the OEMS. In this way, any hydrolysis reactions occurring at room temperature before the start of the experiment were minimized. The temperature chamber was then set to constant temperature hold steps of 10°C, 25°C, 40°C, and 60°C, each for 3 h, as previously described by Metzger et al.³⁸ Finally, the temperature was set to 80°C for 12 h to follow the long-term reactions of LiPF_6 decomposition products.

For electrochemical oxidation measurements, 500 mg carbon black (C65, Timcal, Imerys, Switzerland) was mixed with 500 mg polyvinylidene difluoride (PVDF, Kynar HSV 900, Arkema, France) and 10 mL N-methyl-pyrrolidone (NMP, anhydrous, Sigma-Aldrich, United States) in a planetary mixer (Thinky Corp., USA) for 15 min at 2000 rpm and coated onto a PET separator (FS 24316, Freudenberg, Germany). After drying at 50°C, 15 mm electrodes (loading 1.8 $\text{mg}_\text{C}/\text{cm}^2$) were punched out, dried at 120°C under dynamic vacuum for 12 h, and then transferred to an Ar-filled glove box without exposure to air. The OEMS cell was assembled with a lithium metal counter electrode (450 μm thickness, Rockwood Lithium, USA), the carbon black coating as working electrode, and two 28 mm diameter PET separators (also dried at 120°C) soaked with 150 μL EC + 1.5 M LiPF_6 . Electrochemical measurements were performed by applying a linear sweep from open circuit voltage (~3 V) to 5.5 V at a rate of 0.1 mV/s using a VMP-300 potentiostat (Biologic, France) after an initial 4 h OCV period.

Results

Thermal decomposition of dry and wet LiPF_6 .—To understand which conditions are required to produce PF_5 or POF_3 in quantitative amounts, we first investigate the thermal decomposition of LiPF_6 under dry and wet conditions by TGA-MS. The possible fragments of PF_5 and POF_3 and their corresponding m/z signals are listed in Table I; to evaluate the experiments, we chose $m/z = 107$ as a unique signal for PF_5 and $m/z = 85$ as a unique signal for POF_3 .

Figure 2 shows the TGA-MS run of LiPF_6 with a dry argon gas flow. As a sample transfer from the glove box to the TGA instrument under inert gas was not possible, we added a 1 hour isothermal period

Table I. Mass signals, corresponding mass fragments, and their relative intensity for PF₅ and POF₃. Data taken from the NIST Mass Spectrometry Data Center Ref. 39. The m/z = 107 and 85 signals (highlighted in bold) are unique for PF₅ and POF₃, respectively, and were chosen for the quantitative analysis of the two gases.

Mass signal [m/z]	Relative intensity [%]		Fragment
	PF ₅	POF ₃	
47	-	2.2	PO
50	4.9	4.1	PF
66	-	0.7	POF
69	3.8	16.9	PF ₂
85	-	85.4	POF₂
88	4.4	2.3	PF ₃
104	-	100	POF ₃
107	100	-	PF₅
126	0.8	-	PF ₅

at 120°C (see 10–70 min in Figure 2) to the ramp of 10 K/min from room temperature to 350°C. Any physisorbed trace water from the sample transfer should be removed during this step. In fact, at the initial ramp from room temperature to 120°C (0–10 min), a small amount of water is released (purple line corresponding to m/z = 18 in Figure 2c) concomitant with a slight endothermic heat flux (orange line in Figure 2b). Both sample mass and heat flow remain constant during the end of the isothermal step at 120°C (10–70 min), meaning that all physisorbed water has been desorbed. Once the temperature in the subsequent ramp from 120 to 350°C (70–93 min) reaches 165°C (see red dotted lines), a significant endothermic mass loss of 83% referenced to the original mass occurs (see Figure 2b). At the same time, a peak in the mass trace m/z = 107 (blue line in Figure 2c) is observed. Both the formation of PF₅ and the mass loss of 83% are consistent with the quantitative thermal decomposition of LiPF₆ according to Equation 1. Interestingly, the found decomposition onset is 30–60°C higher compared to previous reports with a comparable setup.^{2,4,5} This could be due to the large sample size (150 mg) used in

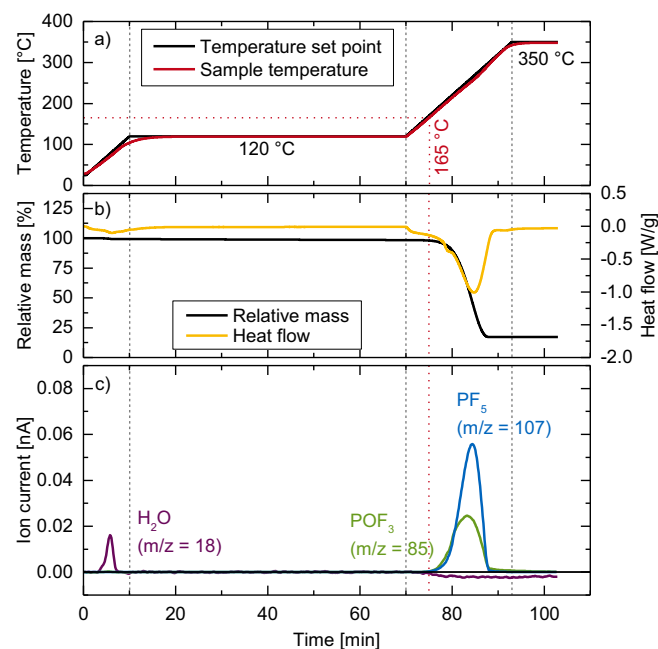


Figure 2. TGA-MS data on the thermal decomposition of dry LiPF₆ under dry argon gas flow: a) temperature protocol setpoints (black) and sample temperature (red); b) mass loss (black) and heat flow (orange); c) mass traces of H₂O (purple), PF₅ (blue), and POF₃ (green).

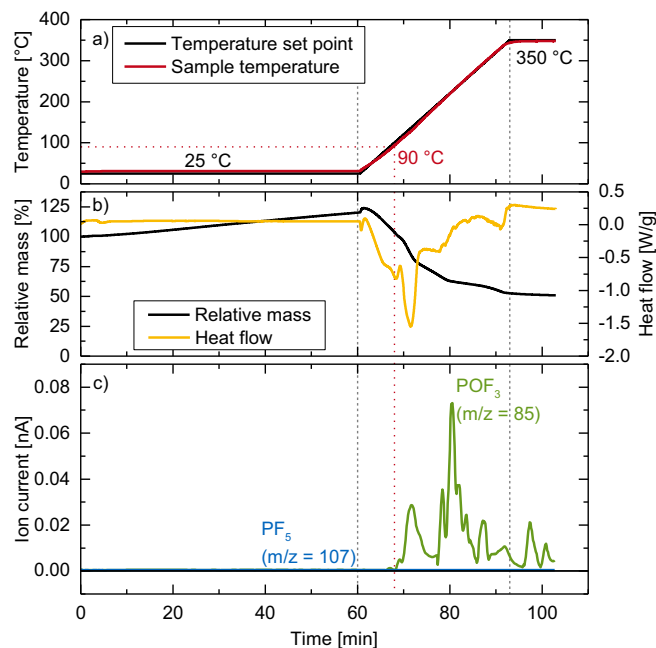


Figure 3. TGA-MS data on the thermal decomposition of dry LiPF₆ under wet argon gas flow: a) temperature protocol setpoint (black) and sample temperature (red); b) mass loss (black) and heat flow (orange); c) mass traces of PF₅ (blue) and POF₃ (green).

the present study, as the resulting higher concentration of PF₅ leads to a thermodynamic shift of the equilibrium toward LiPF₆.^{2,3}

Besides the mass fragments for PF₅, we also observe a signal at m/z = 85 (green line in Figure 2c), which belongs to the POF₂-moiety of POF₃ (see Table I). Furthermore, a slight decrease of the m/z = 18 background can be seen. This suggests that part of the PF₅ reacts with trace water in the instrumentation to form POF₃ (see Equation 4), despite the initial drying step and the use of dry argon (5.0 purity), illustrating the high reactivity of PF₅ with trace water:

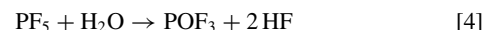


Figure 3 shows the thermal decomposition of LiPF₆ in the presence of water. To achieve a significant “wetting” of the LiPF₆, water-saturated argon gas (dewpoint of ~20°C) is flown over the sample before the temperature ramp is started (0–60 min in Figure 3). During this time, a mass increase of ~24% is observed (black line in Figure 3b), consistent with the weight gain expected for the formation of an adduct with the nominal stoichiometry of LiPF₆ · 2 H₂O (note that in this experiment the background of the H₂O signal (m/z = 18) was too high to observe any changes in the water concentration). It is to note that the sample weight still increases after the initial 60 min, which implies that a further uptake of water would have been possible if the wetting time was extended; however to stay in the same time scale as in the “dry” experiment shown in Figure 2, we limited the pre-wetting in Figure 3 to one hour. The subsequent ramp from room temperature to 350°C (10 K/min) correlates with a mass loss of only 49% referenced to the original LiPF₆ mass, accompanied by a strong endothermic heat flow. This time, no m/z = 107 signal belonging to PF₅ was observed (see blue line in Figure 3c); instead, only POF₃ (m/z = 85) is detected (green line in Figure 3c). Both the mass loss and the POF₃ evolution start around 90°C (see red dotted lines), i.e., 75°C lower compared to the experiment with dry argon gas flow in Figure 2, which is in good agreement with previous reports on the decomposition of intentionally wetted LiPF₆.^{4,5} In contrast to these studies, however, the mass loss in our wet experiment (49%) is much lower than the theoretical mass loss based on a complete conversion to LiF according to Equation 2 (83%). This means that another thermally stable species besides LiF must be produced. It is possible that during

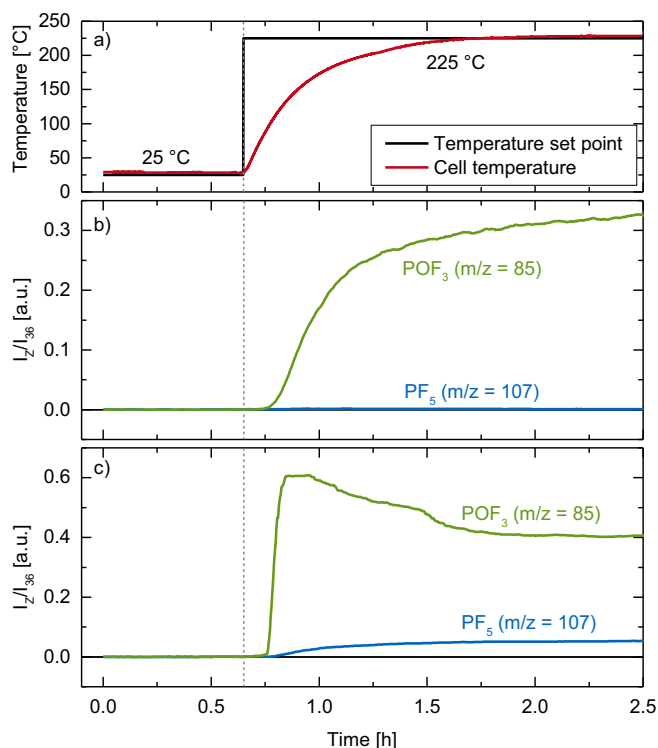
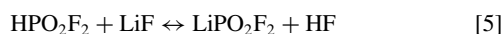


Figure 4. Thermal decomposition experiment with dry LiPF_6 in the modified OEMS cell: a) temperature protocol (black) and cell temperature (red); b) mass traces for POF_3 (green) and PF_5 (blue) from 0.26 mg LiPF_6 without pre-drying of the OEMS inlet; c) mass traces for POF_3 (green) and PF_5 (blue) from 0.37 mg after heating of the OEMS inlet at 120°C under vacuum for 48 h before the start of the experiment.

the course of the TGA experiment, HPO_2F_2 is formed according to Equation 3, which then reacts with remaining LiF in the following equilibrium:⁴⁰



With a melting point of 360°C ,⁴⁰ LiPO_2F_2 will not decompose but contribute to the remaining mass at the end of the TGA experiment; if the conversion of LiPF_6 to LiPO_2F_2 were to be quantitative (the sum of Reactions 2, 3, and 5), the overall mass loss referenced to LiPF_6 would be 29%. Thus, if we assume that LiF and LiPO_2F_2 are the only thermally stable products, the mass loss of 49% would correspond to a composition of 37% LiF and 63% LiPO_2F_2 , based on the molar masses of $M(\text{LiF}) = 25.9 \text{ g/mol}$ and $M(\text{LiPO}_2\text{F}_2) = 107.9 \text{ g/mol}$. However, a detailed chemical analysis of the sample residue would be necessary to verify this hypothesis.

Thermal decomposition of LiPF_6 by OEMS.—To obtain a quantitative relationship between the ion current measured in our OEMS system and the concentration of PF_5 or POF_3 in the cell headspace, we aimed to thermally decompose LiPF_6 inside a cell connected to our OEMS system. Therefore, we assembled a modified OEMS cell which can be heated up to 250°C (see Experimental section and Figure 1 for details). Once the cell would be heated to 225°C , i.e., well above the decomposition temperature of dry LiPF_6 at 165°C (see Figure 2), we expected LiPF_6 to decompose stoichiometrically to PF_5 according to Equation 1. As the expected partial pressure of PF_5 is less than 20 mbar, a shift of the equilibrium to the left of Equation 1 can also be neglected. Figure 4 shows the cell temperature and the characteristic mass traces for PF_5 ($m/z = 107$, blue line in Figure 4b) and POF_3 ($m/z = 85$, green line in Figure 4b) during the decomposition of LiPF_6 under regular dry conditions in our modified OEMS cell. Surprisingly, only mass traces belonging to POF_3 were detected, although all cell

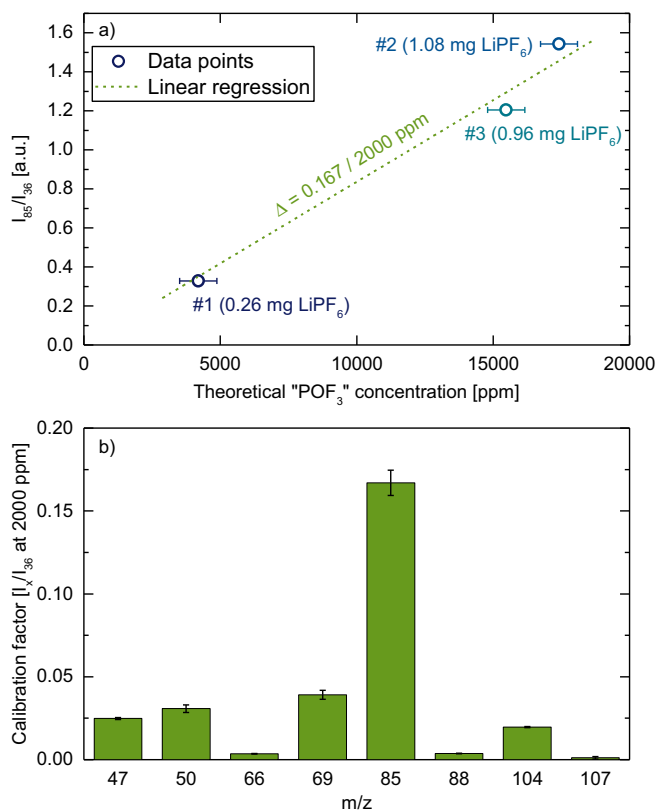


Figure 5. Quantification of the apparent amount of “ POF_3 ” (\equiv sum of $\text{PF}_5 + \text{POF}_3$) from the thermal decomposition of different amounts of dry LiPF_6 at 225°C in the modified OEMS cell. a) Ion current on $m/z = 85$ normalized to $m/z = 36$ vs. the theoretical concentration of “ POF_3 ”; error bars reflect the weighing accuracy of $\pm 0.04 \text{ mg}$. The calibration factor is determined by linear regression between the data points (see green dashed line). b) Calibration factor of “ POF_3 ” normalized to the ^{36}Ar current and to 2000 ppm ($\equiv I_x/I_{36}$ at 2000 ppm “ POF_3 ”) for different m/z channels obtained from linear regression lines, whereby the error bars represent the standard deviation of the determined slopes. Experiments were conducted with 0.26, 0.96, and 1.08 mg dried LiPF_6 .

pieces and the LiPF_6 salt had been dried carefully and the cell assembly was done in an Ar-filled glove box. We repeated the experiment twice with different amounts of LiPF_6 (see Figure 5), which however still did not lead to detectable amounts of PF_5 on the mass channel $m/z = 107$. This clearly suggests that the reaction of PF_5 produced in the cell with either trace water in the tubing of the high-vacuum side of the capillary or with the native oxide on stainless steel surfaces leads to a quantitative conversion to POF_3 .

In order to remove any trace water within the tubing of the high-vacuum side of the OEMS setup, we conducted another experiment where the corrugated tube which connects the OEMS capillary with the mass spectrometer was heated to 120°C for 48 h under ultra-high vacuum (see yellow shaded area in Figure 1), while the modified OEMS cell was already attached with the capillary leak closed but kept near room temperature. This allowed to subsequently perform the LiPF_6 decomposition experiment with the OEMS inlet perfectly dried prior to initiating the thermal decomposition of LiPF_6 in the OEMS cell. The results of this experiment are depicted in Figure 4c. While in this experiment POF_3 is still the by far dominant species (green line, $m/z = 85$), a small amount of PF_5 (blue line, $m/z = 107$) could now be observed. As the partial pressure of PF_5 in the tubes beyond the capillary is $\sim 10^8$ times lower than in the OEMS cell (10^{-5} vs. 10^3 mbar),³⁷ very low amounts of adsorbed water or oxides on the stainless steel tubing surface are obviously sufficient to convert most of the thermally formed PF_5 to POF_3 before it reaches the quadrupole. The apparent decrease of POF_3 in the beginning of the experiment (1.0–1.5 h in Figure 4c) is thus most likely because

Table II. Mass signal channels and calibration factors for the quantification of different gases for our OEMS system. The calibration factors are referenced to the ^{36}Ar signal and normalized to 2000 ppm of the respective gas. Data for H_2 , C_2H_4 , CO , and CO_2 is taken from Ref. 41, whereas the calibration factor for “ POF_3 ” was determined in this work.

Gas	Mass signal m/z	Calibration factor [I_x/I_{36} at 2000 ppm]
H_2	2	0.15
C_2H_4	26	0.38
CO	28	0.64
CO_2	44	0.58
“ POF_3 ”	85	0.17

PF_5 emerging through the capillary reacts with initially present trace amounts of water in the OEMS inlet, leading to a relatively higher POF_3 signal at $m/z = 85$; after depletion of trace water, increasing amounts of PF_5 at $m/z = 107$ can be observed, concomitant with a simultaneously lower POF_3 signal.

This experiment shows that even if PF_5 were to form during conventional OEMS experiments in our setup, it will react with trace water (see Equation 4) and/or surface oxides on the stainless steel tubes within the OEMS inlet and thus will predominantly be detected as POF_3 . Consequently, at least in our OEMS setup, the above experiment shows that we are not able to differentiate between POF_3 and PF_5 in our measurements. For this reason, it is quite likely that at least part of the $m/z = 85$ signals which previously had been assigned to POF_3 by our group^{31,32} may actually have been due to PF_5 . While we do not know whether this artefact might also be occurring with other on-line mass spectrometry systems used for the study of lithium ion batteries, the above experiments certainly suggest that it would be worthwhile to examine the extent of the PF_5 to POF_3 conversion for each system.

Although we cannot distinguish between PF_5 and POF_3 in our OEMS setup, we can still establish an at least semi-quantitative calibration factor for the apparent amount of “ POF_3 ”, representing the sum of $\text{PF}_5 + \text{POF}_3$. This will be done by correlating the amount of thermally decomposed LiPF_6 with the measured ion currents, evaluating the ion currents on all significant mass channels once they stayed constant, which was typically 3–4 h after the start of the heating experiment. We only considered the experiments where all PF_5 was converted to POF_3 (i.e., without heated tubing), as this would be the default case for all OEMS measurements conducted in our laboratory. Figure 5a depicts the ion current on $m/z = 85$ normalized to the ^{36}Ar isotope (I_{85}/I_{36} , y-axis) vs. the theoretical concentration of “ POF_3 ” (referring to apparent POF_3) from the thermal decomposition of LiPF_6 (x-axis) for the three measurements with 0.26, 0.96, and 1.08 mg LiPF_6 (± 0.04 mg, corresponding to 4200, 15500, and 17400 ppm “ POF_3 ”). Calibration factors were determined by the linear regression slope between the three data points shown in Figure 5a, which was then referenced to 2000 ppm “ POF_3 ”. Figure 5b shows the thus determined calibration factors corresponding to 2000 ppm of “ POF_3 ” for the different POF_3 mass channels after normalizing their ion currents to the ion current for ^{36}Ar (I_x/I_{36}) signals. It can be seen that the fragment on $m/z = 85$ (POF_2) has the highest intensity, corresponding to a calibration factor of 0.167 ($\equiv I_{85}/I_{36}$ at 2000 ppm “ POF_3 ”), while the fragments with $m/z = 47$ (PO), $m/z = 50$ (PF), $m/z = 69$ (PF_2), and $m/z = 104$ (POF_3) all have much lower relative intensities. Note that this differs significantly from the fragmentation reported by the National Institute of Standards and Technology (NIST) for the same ionization energy of 70 eV (see Table I),³⁹ which we ascribe to the fact that the ionization chamber in our instrument is heated at 120°C.³⁷ Table II shows the calibration factors for H_2 , C_2H_4 , CO , and CO_2 (determined by purging the cell with a test gas containing 2000 ppm of each species), along with the newly determined calibration factor for “ POF_3 ”, which lies within the same range (0.1–0.7) as the other calibration factors. Thus, when only POF_3 related signals are observed

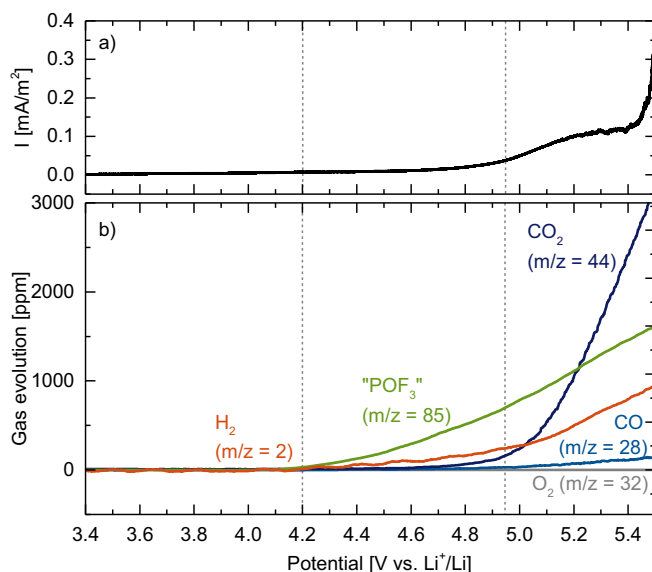


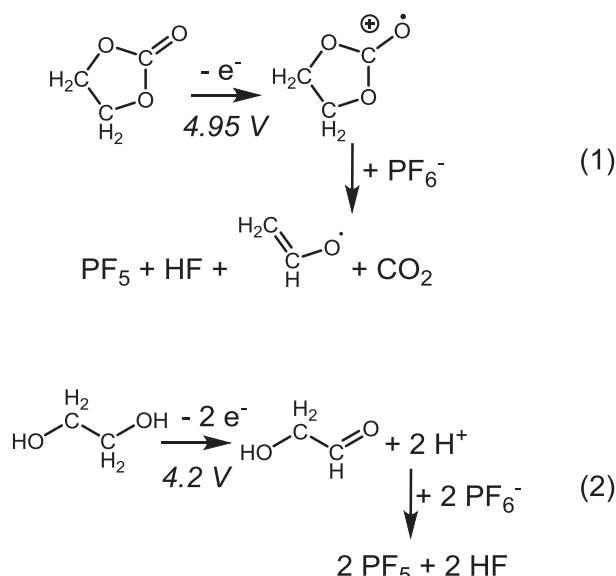
Figure 6. a) Current density and b) gas evolution during a linear oxidative potential scan of a carbon black electrode in EC + 1.5 M LiPF_6 electrolyte. The vertical dashed lines at 4.2 V and 4.95 V mark the approximate onset for the oxidation of ethylene glycol and ethylene carbonate, respectively. The mass signals were converted into concentrations using the calibration factors given in Table II.

in our OEMS setup, the sum of $\text{PF}_5 + \text{POF}_3$ can be quantified, even though their distribution cannot be determined.

Electrochemical oxidation of LiPF_6 electrolyte.—As already mentioned, POF_3 has been observed at high positive potentials on cathode active materials or on carbon electrodes in LiPF_6 -containing electrolytes at room temperature, but its amount has never been quantified.^{29–32} Therefore, we examined the oxidation reactions of an EC + 1.5 M LiPF_6 electrolyte on a carbon black working electrode vs. a lithium counter electrode, focusing on the evolution and quantification of LiPF_6 decomposition species. Figure 6 shows the current density (a) and the gas evolution (b) during a linear potential sweep from OCV to 5.5 V vs. Li^+/Li , whereby all signals have been quantified using the calibration factors given in Table II (note that only signals related to POF_3 were observed, so that a quantification of “ POF_3 ” is possible). Around 4.2 V vs. Li^+/Li , the evolution of “ POF_3 ” (i.e., $\text{POF}_3 + \text{PF}_5$, green line in Figure 6b) sets in, together with the formation of H_2 (orange line in Figure 6b). Starting at 4.95 V vs. Li^+/Li , a significant oxidation current (black line in Figure 6a) and the simultaneous evolution of CO_2 (dark blue line in Figure 6b) are observed, as previously reported from the oxidation of EC-based electrolytes,^{35,42,43} along with an enhanced formation of H_2 and “ POF_3 ”. Density functional theory calculations by Borodin et al.²⁴ and Li et al.²⁵ have pointed out that upon EC oxidation (i.e., after the first electron transfer), the abstraction of a proton by a neighboring PF_6^- anion would occur in LiPF_6 -based electrolytes, leading to HF and PF_5 formation, and ultimately to the release of CO_2 and a vinyl alcoholate radical (see reaction pathway (1) in Scheme 1). The produced PF_5 would then be detected as “ POF_3 ” in our OEMS setup, while the reduction of HF at the Li metal counter electrode²¹ would result in the evolution of H_2 :



However, contrary to this mechanism, the evolution of H_2 and “ POF_3 ” between 4.2–4.95 V occurs without the simultaneous formation of CO_2 , suggesting that a different process is taking place in this potential range. As trace amounts of water gradually react with EC to form ethylene glycol,^{22,38} the latter is a likely impurity in EC present at ppm levels (unfortunately below the NMR detection level).



Scheme 1. (1) Oxidation mechanism of EC in the presence of PF_6^- , based on Refs. 25,48. (2) Oxidation of ethylene glycol according to Refs. 44,45.

The oxidation of ethylene glycol in aqueous electrolytes leads to the formation of protons,^{44,45} which could also react with PF_6^- to HF and PF_5 (see pathway (2) in Scheme 1). In fact, the amounts of H_2 and “ POF_3 ” evolved up to 4.95 V would correspond to the oxidation of ~20 ppm ethylene glycol, which is a probable concentration for this impurity.

For either mechanism (1) or (2) in Scheme 1, the predicted molar ratio of PF_5/HF would be 1/1, so that the experimental molar ratio of “ POF_3 ” (representing PF_5) and H_2 should be 2/1 (since the reduction of 1 HF produces 0.5 H_2 ; see reaction 6). This is in good agreement with the data in Figure 6, where the amount of H_2 is indeed about 50% compared to that of “ POF_3 ”. In summary, the data shown in Figure 6 lead us to the following hypothesis: a) protons or acidic species from EC or ethylene glycol impurity oxidation lead to a fast dissociation of PF_6^- to HF and PF_5 already at room temperature, and b) POF_3 observed at oxidative potentials in OEMS experiments on a carbon black electrode is in fact mainly PF_5 rather than POF_3 as we had assumed previously.³²

To verify whether the formation of acidic species (e.g., HF or H^+) can lead to a significant dissociation of LiPF_6 at room temperature within the timescale of an OEMS experiment, we investigated the reaction of an EC + 1.5 M LiPF_6 electrolyte with 5000 ppm methanesulfonic acid (MSA, $\text{pK}_\text{A} = 8.3$ in propylene carbonate).⁴⁶ With this electrolyte, we performed a similar experiment as described by Metzger et al.,³⁸ namely monitoring the gas evolution while gradually increasing the temperature of the electrolyte from 10°C to 80°C. Note that for this type of experimental procedure no active electrodes or lithium are present, so that it only probes purely chemical reactions of the electrolyte. The cell temperature and the resulting gas evolution are shown in Figures 7a and 7b. A significant amount of “ POF_3 ” ($\text{POF}_3 + \text{PF}_5$, green line in Figure 7b) is already formed at 25°C, leveling off at ~1000 ppm. Going to higher temperatures, the overall “ POF_3 ” concentration increases further (from ~3000 ppm at 40°C to ~11000 ppm at 80°C); however, in contrast to the constant steady-state concentration reached at 25°C, at higher temperatures the “ POF_3 ” evolution reaches a maximum within ~1 hour at the respective temperature step and thereafter decreases gradually (a more detailed discussion follows below). CO_2 (navy line in Figure 7b) is only evolved above 60°C, but remains at a much lower concentration compared to “ POF_3 ”; at 80°C, the amount of CO_2 increases continuously.

Clearly, the detection of “ POF_3 ” at 25°C in the MSA-containing electrolyte demonstrates that protons can rapidly and substantially

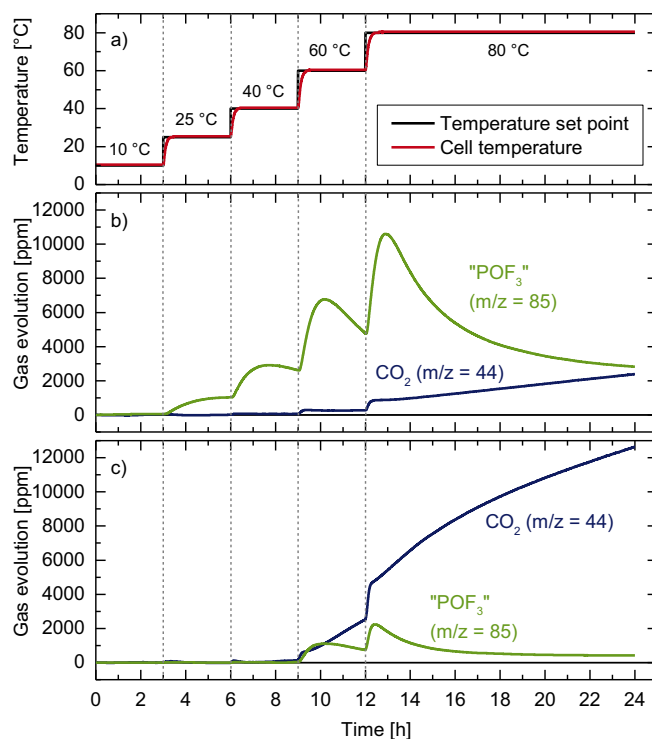


Figure 7. Temperature induced decomposition of EC + 1.5 M LiPF_6 electrolyte, measured in an OEMS cell by stepping the temperature from 10 to 80°C: a) temperature protocol (black) and cell temperature (red); b) “ POF_3 ” and CO_2 evolution in the presence of 5000 ppm methanesulfonic acid; c) “ POF_3 ” and CO_2 evolution in the presence of 5000 ppm water. The mass signals were converted into concentrations using the calibration factors given in Table II.

shift the dissociation of PF_6^- toward PF_5 and HF (see Equation 7). This is in good agreement with Freire et al.,⁴⁷ who found that PF_6^- -based ionic liquids hydrolyze in aqueous solutions with $\text{pH} = 3$ at room temperature, but not under neutral conditions.



At 25°C, the establishment of a steady-state “ POF_3 ” concentration and the absence of CO_2 indicate that at this low temperature no PF_5 is consumed in a reaction with the electrolyte or that this reaction is too slow to be detected within the timescale of the OEMS experiment (~3 h). In contrast, the decreasing concentrations of “ POF_3 ” during the second half of the temperature step at >40°C are likely a result of PF_5 reacting with the electrolyte to oligomers⁸ or alkyl fluorophosphates.^{2,7,9–11} As the rate of chemical PF_5 consumption will depend on both temperature and the concentration of PF_5 dissolved in the electrolyte (which is by Henry’s law proportional to the amount of PF_5 in the headspace of the cell), a faster consumption of POF_3 is expected at 60°C and 80°C, where both temperature and overall “ POF_3 ” concentration are higher. Additionally, a stepwise increase of the temperature can cause the desorption of PF_5 or POF_3 from the inner surface of the steel tubing, which leads to a peak in the detected “ POF_3 ” concentration at the initial phase of each temperature step.

Interestingly, the extent of CO_2 formation in Figure 7b is very similar to that observed with an EC/ LiClO_4 electrolyte with < 20 ppm water reported by Metzger et al.³⁸ (see Figure 1 in Ref. 36), which means that the direct reaction of PF_5 with EC does not generate additional CO_2 . For comparison, Figure 7c shows the same experiment with 5000 ppm of added water instead of MSA. Up to 40°C, neither “ POF_3 ” (green line in Figure 7c) nor CO_2 (navy line in Figure 7c) are observed. Only at temperatures 60°C and above, low amounts (< 2000 ppm) of “ POF_3 ” are detected, in this case likely due to the formation of POF_3 rather than PF_5 (this, however, cannot be discerned

in our OEMS experiments). At the same time, the CO₂ concentration increases linearly at 60°C and 80°C to above 12000 ppm, in agreement with previous experiments on the water-driven hydrolysis of EC in LiClO₄-based electrolytes.³⁸

Comparing the results of Figures 7b and 7c, it becomes apparent that only highly acidic species can trigger the formation of PF₅ at room temperature. Accordingly, the oxidation of an EC/LiPF₆ electrolyte (see Figure 6) must be leading to the formation of species which act as proton donors or Brønsted acids, which is consistent with the mechanism proposed by Borodin et al.²⁴ and Li et al.²⁵ (see Scheme 1). Furthermore, the release of PF₅ might explain the strong temperature dependence of the oxidative stability of LiPF₆-based electrolytes.³³ In this context, the use of proton-scavenging additives, i.e., bases,^{9,18} could be a successful strategy to prevent the dissociation of PF₆⁻ at high potentials, thereby suppressing HF formation and electrolyte degradation.

Conclusions

In this work, we have investigated the thermal and oxidative decomposition reactions of LiPF₆ in Li-ion battery electrolytes. The decomposition onsets of dry and wet LiPF₆ were determined by TGA-MS to be 165°C and 90°C, respectively. While dry LiPF₆ is known to decompose to PF₅ and HF, some POF₃ was observed by TGA-MS due to trace water impurities in the dry argon carrier gas. On the other hand, the thermal decomposition of wet LiPF₆ in water-saturated argon carrier gas resulted in the formation of POF₃, HF, and LiPO₂F₂.

Analyzing the thermal decomposition of dry LiPF₆ in the on-line electrochemical mass spectrometry (OEMS) system developed by our group, we found that all of the thermally released PF₅ is actually detected as POF₃ (*m/z* = 85) under standard experimental conditions, so that it is not possible to distinguish between PF₅ and POF₃ in OEMS experiments. As this might also apply to other on-line mass spectrometry systems developed for the study of lithium ion batteries, the calibration of these systems for their ability to distinguish between these two gases is advisable. This information is crucial for mechanistic studies, as otherwise experimental on-line mass spectrometry data may be mis-interpreted. However, by decomposing specific amounts of LiPF₆, we could establish a calibration factor for the sum of POF₃ + PF₅ (referred to as “POF₃”), allowing us to quantify the amount of the sum of both gases in our OEMS experiments.

Subsequently, we investigated the oxidation of an EC/LiPF₆ electrolyte on a carbon black electrode. Next to CO₂, we observed “POF₃” and H₂ in a ratio of 2:1, suggesting that PF₆⁻ is decomposed by protic species formed during electrolyte oxidation to PF₅ and HF, whereby the latter is reduced to H₂ on the lithium counter electrode. OEMS studies on the thermal stability of EC/LiPF₆ electrolytes with intentionally added protons (in the form of methanesulfonic acid) or water showed that protons can indeed trigger the formation of PF₅ at room temperature, whereas the reaction of LiPF₆ with water is too slow at 25°C to be sensed by OEMS. Hence, we confirmed that the oxidation products of battery electrolytes act as Brønsted acids and trigger the decomposition of PF₆⁻ to PF₅ and HF already at room temperature.

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References

- K. Xu, *Chem. Rev.*, **104**, 4303 (2004).
- B. Ravdel, D. P. Abraham, R. Gitzendanner, J. F. DiCarlo, B. L. Lucht, and C. L. Campion, *J. Power Sources*, **119–121**, 805 (2003).
- E. Zinigrad, L. Larush-Asraf, J. S. Gnanaraj, M. Sprecher, and D. Aurbach, *Thermochim. Acta*, **438**, 184 (2005).
- M. D. S. Lekgoathi, B. M. Vilakazi, J. B. Wagener, J. P. Le Roux, and D. Moolman, *J. Fluor. Chem.*, **149**, 53 (2013).
- H. Yang, G. V. Zhuang, and P. N. Ross, *J. Power Sources*, **161**, 573 (2006).
- M. Pyschik, M. Winter, and S. Nowak, *Separations*, **4** (2017).
- V. Kraft, W. Weber, M. Grütze, M. Winter, and S. Nowak, *RSC Adv.*, **5**, 80150 (2015).
- S. E. Sloop, J. K. Pugh, J. B. Kerr, K. Kinoshita, S. Wang, J. B. Kerr, and K. Kinoshita, *Electrochem. Solid-State Lett.*, **4**, A42 (2001).
- C. L. Campion, W. Li, W. B. Euler, B. L. Lucht, B. Ravdel, J. F. DiCarlo, R. Gitzendanner, and D. P. Abraham, *Electrochem. Solid-State Lett.*, **7**, A194 (2004).
- C. L. Campion, W. Li, and B. L. Lucht, *J. Electrochem. Soc.*, **152**, A2327 (2005).
- S. L. Guillot, A. Pe, M. L. Usrey, and R. J. Hamers, *J. Electrochem. Soc.*, **164**, 1907 (2017).
- V. Kraft, W. Weber, B. Streipert, R. Wagner, C. Schultz, M. Winter, and S. Nowak, *RSC Adv.*, **6**, 8 (2015).
- S. Wilken, M. Treskow, J. Scheers, P. Johansson, and P. Jacobsson, *RSC Adv.*, **3**, 16359 (2013).
- P. Handel, G. Fauler, K. Kapper, M. Schmuck, C. Stangl, R. Fischer, F. Uhligh, and S. Koller, *J. Power Sources*, **267**, 255 (2014).
- S. F. Lux, I. T. Lucas, E. Pollak, S. Passerini, M. Winter, and R. Kostecki, *Electrochem. Commun.*, **14**, 47 (2012).
- S. F. Lux, J. Chevalier, I. T. Lucas, and R. Kostecki, *ECS Electrochem. Lett.*, **2**, A121 (2013).
- K. Tasaki, K. Kanda, S. Nakamura, and M. Ue, *J. Electrochem. Soc.*, **150**, A1628 (2003).
- A. V. Plakhotnyk, L. Ernst, and R. Schmutzler, *J. Fluor. Chem.*, **126**, 27 (2005).
- R. Yazami and A. Martinet, in *Fluorinated Materials for Energy Conversion*, T. Nakajima and H. Groult, Editors, p. 173, Elsevier Ltd (2005).
- T. Kawamura, S. Okada, and J. Yamaki, *J. Power Sources*, **156**, 547 (2006).
- D. Strmcnik, I. E. Castelli, J. G. Connell, D. Haering, M. Zorko, P. Martins, P. P. Lopes, B. Genorio, T. Østergaard, H. A. Gasteiger, F. Maglia, B. K. Antonopoulos, V. R. Stamenkovic, J. Rossmeisl, and N. M. Markovic, *Nat. Catal.*, **1**, 255 (2018).
- U. Heider, R. Oesten, and M. Jungnitz, *J. Power Sources*, **7** (1999).
- V. Kraft, M. Grütze, W. Weber, M. Winter, and S. Nowak, *J. Chromatogr. A*, **1354**, 92 (2014).
- O. Borodin, W. Behl, and T. R. Jow, *J. Phys. Chem. C*, **117**, 8661 (2013).
- T. Li, L. Xing, W. Li, Y. Wang, M. Xu, F. Gu, and S. Hu, *J. Power Sources*, **244**, 668 (2013).
- J. L. Tebbe, T. F. Fuerst, and C. B. Musgrave, *ACS Appl. Mater. Interfaces*, **8**, 26664 (2016).
- W. Weber, R. Wagner, B. Streipert, V. Kraft, M. Winter, and S. Nowak, *J. Power Sources*, **306**, 193 (2016).
- D. Streich, A. Guéguen, M. A. Mendez, F. F. Chesneau, P. Novák, and E. J. Berg, *J. Electrochem. Soc.*, **163**, A964 (2016).
- A. Guéguen, D. Streich, M. He, M. A. Mendez, F. F. Chesneau, P. Novák, and E. J. Berg, *J. Electrochem. Soc.*, **163**, A1095 (2016).
- M. He, L. Boulet-Roblin, P. Borel, C. Tessier, P. Novák, C. Villevieille, and E. J. Berg, *J. Electrochem. Soc.*, **163**, A83 (2016).
- A. Freiberg, M. Metzger, D. Haering, S. Bretzke, S. Puravankara, T. Nilges, C. Stinner, C. Marino, and H. A. Gasteiger, *J. Electrochem. Soc.*, **161**, A2255 (2014).
- M. Metzger, J. Sicklinger, D. Haering, C. Kavakli, C. Stinner, C. Marino, and H. A. Gasteiger, *J. Electrochem. Soc.*, **162**, A1227 (2015).
- M. Metzger, P. Walke, S. Solchenbach, G. Salitra, D. Aurbach, and H. A. Gasteiger, *Manuscript in Preparation*.
- R. Jung, M. Metzger, F. Maglia, C. Stinner, and H. A. Gasteiger, *J. Electrochem. Soc.*, **164**, A1361 (2017).
- R. Jung, M. Metzger, F. Maglia, C. Stinner, and H. A. Gasteiger, *J. Phys. Chem. Lett.*, **8**, 4820 (2017).
- J. Wandt, A. Freiberg, A. Ogronnik, and H. A. Gasteiger, *Mater. Today*, in press (2018).
- N. Tsiouvaras, S. Meini, I. Buchberger, and H. A. Gasteiger, *J. Electrochem. Soc.*, **160**, A471 (2013).
- M. Metzger, B. Strehle, S. Solchenbach, and H. A. Gasteiger, *J. Electrochem. Soc.*, **163**, A1219 (2016).
- <http://webbook.nist.gov>.
- T. H. Tan, *Master's Thesis*, The University of British Columbia (1970).
- B. Strehle, S. Solchenbach, M. Metzger, K. U. Schwenke, and H. A. Gasteiger, *J. Electrochem. Soc.*, **164**, A2513 (2017).
- D. Pritzl, S. Solchenbach, M. Wetjen, and H. A. Gasteiger, *J. Electrochem. Soc.*, **164**, A2625 (2017).
- M. Metzger, B. Strehle, S. Solchenbach, and H. A. Gasteiger, *J. Electrochem. Soc.*, **163**, A798 (2016).
- G. Horanyi, V. E. Kazarinov, Y. B. Vassiliev, and V. N. Andreev, *J. Electroanal. Chem.*, **147**, 263 (1983).
- B. Wieland, J. P. Lancaster, C. S. Hoaglund, P. Holota, and W. J. Tornquist, *Langmuir*, **12**, 2594 (1996).
- L. Xing, O. Borodin, G. D. Smith, and W. Li, *J. Phys. Chem. A*, **115**, 13896 (2011).
- K. Izutsu, M. Kolthoff, T. Fujinaga, M. Hattori, and K. Chantooni, *Anal. Chem.*, **49**, 503 (1977).
- M. G. Freire, C. M. S. S. Neves, I. M. Marrucho, and A. P. Coutinho, *J. Phys. Chem. A*, **2**, 3744 (2010).